

NUMERICAL ANALYSIS OF THE VERTEX MODELS FOR SIMULATING GRAIN BOUNDARY NETWORKS*

C. E. TORRES[†], M. EMELIANENKO[‡], D. GOLOVATY[§], D. KINDERLEHRER[¶],
AND S. TA'ASAN[¶]

Abstract. Polycrystalline materials undergoing coarsening can be represented as evolving networks of grain boundaries, whose statistical characteristics describe macroscopic properties. The formation of various statistical distributions is extremely complex and is strongly influenced by topological changes in the network. This work is an attempt to elucidate the role of these changes by conducting a thorough numerical investigation of one of the simplest types of grain growth simulation models, the vertex model. While having obvious limitations in terms of its ability to represent realistic systems, the vertex model enables full control over topological transitions and retains essential geometric features of the network. We formulate a self-consistent vertex model and investigate the role of microscopic parameters on mesoscale network behavior. This study sheds light on several important questions, such as how statistics are affected by the choice of temporal and spatial resolution and rules governing topological changes. Statistical analysis of the data produced by the simulation is performed for both isotropic and anisotropic grain boundary energies.

Key words. vertex model, topological transitions, grain growth, polycrystalline materials

AMS subject classifications. 37M05, 35Q80, 93E03

DOI. 10.1137/140999232

1. Introduction. Polycrystalline materials such as metals and ceramics are composed of single crystallites, called grains, separated by their boundaries, called grain boundaries. The orientations, shapes, and arrangements of the grains have a direct relationship to macroscopic materials properties. For example, the presence of grain boundaries decreases thermal and electrical conductivity that affects the performance of chips in microprocessors. Grain boundaries disrupt motion of dislocations through a material, so reducing crystallite size is a common way to improve strength and fracture toughness in structures, as described by the Hall–Petch relationship [1].

The grain and grain boundary configuration, or the microstructure of a material, is determined by a variety of factors, such as history of deformation, phase transitions, and heat treatment. In this paper, we are primarily interested in the process of microstructural relaxation known as coarsening. Evolution of the grain boundary network during coarsening is driven by the tendency of the system to reduce its total grain boundary surface energy that results in growth of some grains at the expense

*Received by the editors December 8, 2014; accepted for publication (in revised form) February 19, 2015; published electronically April 14, 2015.

<http://www.siam.org/journals/siap/75-2/99923.html>

[†]Department of Mathematical Sciences, George Mason University, Fairfax VA 22015. Current address: Departamento de Informática and Centro Científico-Tecnológico de Valparaíso, Universidad Técnica Federico Santa María, Casilla 110-V, Valparaíso, Chile (ctorres@inf.utfsm.cl). This author's work was supported in part by NSF grant DMS-1056821.

[‡]Department of Mathematical Sciences, George Mason University, Fairfax, VA 22015 (memelian@gmu.edu). This author's work was supported in part by NSF grant DMS-1056821.

[§]Department of Mathematics, University of Akron, Akron, OH 44325 (dmitry@uakron.edu). This author's work was supported in part by NSF grant DMS-1009849.

[¶]Department of Mathematical Sciences, Carnegie Mellon University, Pittsburgh, PA 15213 (davidk@andrew.cmu.edu, shlomo@andrew.cmu.edu). The fourth author's work was supported in part by NSF grants DMS-0806703, DMS-0635983, and OISE-0967140. The fifth author's work was supported in part by NSF grant DMS-1216433.

of others, as well as in disappearance and nucleation of both small grains and grain boundaries. In this process the average grain size increases, while the total surface area of the grain boundaries decreases. As it evolves, the grain boundary network begins to exhibit stable, self-similar statistical features that can be described by a finite number of time-dependent parameters. These parameters, in turn, can be used as continuum descriptors of the network.

One of the principal goals of mathematical modeling of polycrystalline materials is to understand how the statistics of an evolving grain boundary network depend on the set of laws that govern the dynamics of the network at the microscopic scale. Here the laws in question describe the motion of grain boundaries and their junctions, as well as the criteria for nucleation and disappearance of grains and grain boundaries. Although these laws have been known for quite some time, their precise role in the development of macroscopic properties of the network is still not fully clear. A possible way to establish a connection between the statistical features of the network and the evolution of individual grain boundaries involves numerical experimentation using large-scale simulation models. As a starting point, it is therefore desirable to consider models that are as simple and as computationally inexpensive as possible yet that preserve the essential characteristics of the original grain boundary network. In what follows we concentrate on two-dimensional polycrystalline materials that can be thought of as, e.g., cross sections of systems of columnar grains in aluminum films [2].

Since the motion of grain boundaries is controlled by surface energy, it is normally modeled within the framework of curvature-driven growth. Under certain assumptions on relative mobilities of the boundaries and their junctions, it is possible to assume that the boundaries remain straight during the evolution so that the changes within the network can be described solely in terms of motion of the junctions, or vertices, of the graph formed by the boundaries. Note that for isotropic surface energies, only the junctions between three grain boundaries, as opposed to four or five, for instance, are stable. Vertex models that discount grain boundary motion in favor of triple junction motion are used extensively, both due to their computational simplicity in handling extremely large scale networks and for the purpose of isolating properties local to triple junctions, for example, triple junction drag. Following the pioneering works of Fullman [3] in the 1950s and Frost et al. in the 1980s [4], a number of extensions of the original algorithm have been proposed [5, 6, 7, 8, 9, 10, 11]. The vertex models have been able to reproduce many characteristic features of the cellular pattern growth in foams [12, 13] and to some extent in polycrystals [14]. They have high flexibility, which motivates continued interest in their use despite the existence of more sophisticated numerical codes. The vertex model approach recently has been applied to the recrystallization of ferritic stainless steels [15] and, more generally, to grain growth [14, 16]. It has also been extensively used to validate topological theories of grain growth and Zener pinning [17, 18]. A comprehensive review of the relevant literature can be found in the chapter dedicated to vertex models in [19].

In this paper we develop a numerical algorithm for a version of a simple vertex model originally proposed by Nagai et al. in [5]. Our main aims are to derive the set of rules for topological transitions within the network that are consistent with continuous evolution of vertices as well as to control the stability and accuracy of the code. This is done in order to eliminate numerical issues from the investigation of the role that various model parameters play in the development of statistical features. We demonstrate that, although simple, our model results in rich statistics that are

reminiscent of what is observed in experiments and more sophisticated simulations. We emphasize, however, that our main motivation is not to replicate or explain experimental observations, but to ensure the correct characterization of the complexity of network behavior. In a subsequent publication we will investigate the formation of statistics using numerical experimentation with the vertex model developed here. This investigation will expand on our prior studies of a one-dimensional model [20, 21, 22].

The paper is organized as follows. In section 2, we start by formulating a general energy-based model of an evolving grain boundary network. We formally demonstrate that this model reduces to a vertex model by assuming that the mobility of the vertices is much lower than the mobility of grain boundaries evolving via curvature-driven motion. Next, in sections 3 and 4 we use semirigorous analysis of vertex dynamics to derive the set of neighbor switching rules as well as estimates of vertex collision times. Stability analysis of the explicit numerical scheme used in the main algorithm is given in section 5. The full description of the algorithm appears in section 6, followed by the numerical results in section 8. We begin this section by testing the numerical procedure for accuracy and convergence. The procedure is then employed to simulate coarsening of grain boundary networks containing a large number of grains. The geometry of configurations that develop in these simulations is described using the standard statistical measures for characterizing grain growth. These include distributions of relative areas of grains, dihedral angle, and number of sides, among others. We are able to confirm spatiotemporal stability of the distributions that emerge in a network evolving via our numerical algorithm. We find that the distributions are essentially independent of the level of numerical resolution as the network appears to pass through the sequence of similar states, possibly at different rates. While mesoscopically the model is insensitive to various modifications, including the rules governing topological changes, the microscopic features of the network tend to differ with the scenario.

2. Vertex model formalism. Let us define the configuration and establish the law of evolution for our network. Suppose that a rectangular domain $R \subset \mathbb{R}^2$ contains a set Γ of $K > 0$ smooth curves $\Gamma_k := \{\mathbf{x} = \boldsymbol{\xi}_k(s), 0 \leq s \leq L_k\}$, $k = 1, \dots, K$, that we will call *grain boundaries*, with $L_k > 0$ being the length of the k th boundary. On a grain boundary curve Γ_k , one can define an orthogonal frame $\{\mathbf{b}_k, \mathbf{n}_k\}$, where

$$\mathbf{b}_k = \frac{d\boldsymbol{\xi}_k}{ds} / \left| \frac{d\boldsymbol{\xi}_k}{ds} \right| \quad \text{and} \quad \mathbf{n}_k = \frac{d\mathbf{b}_k}{ds} / \left| \frac{d\mathbf{b}_k}{ds} \right|.$$

Assuming periodic boundary conditions on ∂R , all grain boundaries can terminate only at junctions with other boundaries. We denote the set of all junctions in R by $\mathbf{X} := \{\mathbf{x}_1^{n_1}, \mathbf{x}_2^{n_2}, \dots, \mathbf{x}_M^{n_M}\}$, where the number of junctions $M \in \mathbb{N}$. Here an n -tuple junction \mathbf{x}_m^n is a terminal point of n grain boundaries $\Gamma_{j_1}, \Gamma_{j_2}, \dots, \Gamma_{j_n}$, for some $j_1, j_2, \dots, j_n \in \{1, \dots, K\}$. In the simplest and most commonly studied type of a grain boundary network, the numbers $n_1 = n_2 = \dots = n_M = 3$, i.e., all elements of \mathbf{X} are *triple junctions*.

The grain boundaries contained in Γ subdivide the domain R into N disjoint regions $\{\Sigma_1, \dots, \Sigma_N\} =: \Sigma$, called *grains*. With each grain $\Sigma_l \in \Sigma$, $l = 1, \dots, N$, we associate an *orientation* $\alpha_l \in [0, 2\pi)$ and the set of grain boundaries $\partial\Sigma_l = \{\Gamma_{k_1}, \dots, \Gamma_{k_l}\}$ that enclose Σ_l . Likewise, for each grain boundary Γ_k , $k = 1, \dots, K$, there are exactly two grains $\Sigma_{l_1(k)}$ and $\Sigma_{l_2(k)}$ that are separated by Γ_k .

The *grain misorientation* parameter $\Delta\alpha_k$ is defined as $\Delta\alpha_k := \alpha_{l_2(k)} - \alpha_{l_1(k)}$, where $k = 1, \dots, K$. The *grain boundary energy*, γ_k , will be assumed to depend only on misorientation, i.e., $\gamma_k = \gamma(\Delta\alpha_k)$ for every $k = 1, \dots, K$ and some given function $\gamma : \mathbb{R} \rightarrow \mathbb{R}$. The function γ is even and periodic with a period that depends on the symmetries of crystalline lattices of neighboring grains.

We will assume that the grain boundary network evolves in time via simultaneous motion of both the grain boundaries and their junctions. In the course of this motion, some grains grow and some shrink. Once the length of a grain boundary or the area of a grain decreases to zero, we will say that a component of the network has disappeared as a result of a *topological transition*. In order to describe the evolution of the grain boundary network, both the laws of continuous motion of the boundaries and junctions as well as the rules governing the topological transitions must be specified. From now on we will assume that the sets Γ, \mathbf{X} , and Σ depend on time $t > 0$.

We begin the discussion of the network dynamics by considering a period of time $[t_0, t_0 + T]$ during which no topological transitions occur. Introduce the total energy of the network

$$E(t) = \sum_{k=1}^K \int_0^{L_k} \gamma(\Delta\alpha_k) |\mathbf{l}_k(s, t)| ds, \quad \mathbf{l}_k = \frac{d\boldsymbol{\xi}_k}{ds},$$

where all curves at t_0 are assumed to be parametrized with respect to their arc length and L_k is the length of Γ_k at the time t_0 . Denoting $\gamma_k = \gamma(\Delta\alpha_k)$, we obtain

$$\frac{d}{dt} E(t) = \sum_{k=1}^K \int_0^{L_k} \gamma_k \frac{\mathbf{l}_k}{|\mathbf{l}_k|} \cdot \frac{\partial \mathbf{l}_k}{\partial t} ds = \sum_{k=1}^K \int_0^{L_k} \gamma_k \mathbf{b}_k \cdot \frac{\partial \mathbf{v}_k}{\partial s} ds = \sum_{k=1}^K \int_0^{L_k} \mathbf{T}_k \cdot \frac{\partial \mathbf{v}_k}{\partial s} ds,$$

where $\mathbf{T}_k = \gamma_k \mathbf{b}_k$ denotes the capillary force, also called the line tension. Further, $\mathbf{v}_k(s, t)$ denotes the velocity of the material point s on the curve Γ_k at the time t so that

$$\frac{\partial \mathbf{l}_k}{\partial t} = \frac{\partial}{\partial t} \left(\frac{\partial \boldsymbol{\xi}_k}{\partial s} \right) = \frac{\partial}{\partial s} \left(\frac{\partial \boldsymbol{\xi}_k}{\partial t} \right) = \frac{\partial \mathbf{v}_k}{\partial s}.$$

Integrating by parts and using the Frenet formula $\frac{\partial \mathbf{b}_k}{\partial s} = \kappa_k |\mathbf{l}_k| \mathbf{n}_k$ we obtain

$$(2.1) \quad \frac{d}{dt} E(t) = - \sum_{k=1}^K \int_0^{L_k} \kappa_k V_k |\mathbf{l}_k| ds - \sum_{m=1}^M \mathbf{v}_m \cdot \sum_{l=1}^{n_m} \mathbf{T}_{m,l},$$

where $\mathbf{T}_{m,l}$ is the capillary force along the grain boundary Γ_{j_l} that ends at the triple junction $\mathbf{x}_m^{n_m}$. Further, κ_k and $V_k = \mathbf{v}_k \cdot \mathbf{n}_k$ are the curvature and the normal velocity of Γ_k , respectively.

The simplest framework to enforce energy dissipation is to assume that the grain boundaries and their junctions follow a version of gradient flow dynamics. Then the normal velocity of the boundary Γ_k and the velocity $\mathbf{v}_m := \frac{d}{dt} \mathbf{x}_m^{n_m}$ of the triple junction $\mathbf{x}_m^{n_m}$ can be written as

$$(2.2) \quad V_k = \mu_k \kappa_k$$

and

$$(2.3) \quad \mathbf{v}_m = \lambda_m \sum_{l=1}^{n_m} \mathbf{T}_{m,l},$$

respectively. Here $\mu_k > 0$ is the mobility of Γ_k and $\lambda_m > 0$ is the mobility of $\mathbf{x}_m^{n_m}$. Then, using (2.1), we have

$$(2.4) \quad \frac{d}{dt}E(t) = - \sum_{k=1}^K \mu_k \int_0^{L_k} \kappa_k^2 |\mathbf{l}_k| ds - \sum_{m=1}^M \lambda_m \left| \sum_{l=1}^{n_m} \mathbf{T}_{m,l} \right|^2 \leq 0.$$

Equations (2.2) and (2.3) are known as the Mullins equation [23, 24] and a variation of the Herring condition [25], respectively; cf. also [26].

When the grain boundary mobility is much higher than that of triple junctions, the grain boundaries Γ_k are essentially straight lines throughout the evolution. The precise asymptotic reduction, which we will not treat here, requires boundary layer analysis near the junctions [27]. Hence the dynamics of the grain boundary network is completely determined by the motion of the triple junctions via the law (2.3) which relates the velocity of a triple junction to the sum of capillary forces acting on it. This reduced model is known as a *vertex model* in the literature and is a subject of study in this work.

In what follows, we will set $\lambda_m = 1$ for all $m = 1, \dots, M$ and, unless noted otherwise, assume that the grain boundary energy is only weakly anisotropic, i.e., $\gamma(\Delta\alpha) = 1 + \varepsilon f(\Delta\alpha)$, where $\varepsilon > 0$ is small. As we will soon see, this assumption ensures that all grain boundary junctions are, in fact, triple junctions and $n_m = 3$ for all $m = 1, \dots, M$. Thus we can refer to triple junctions simply as \mathbf{x}_m , $m = 1, \dots, M$, by dropping the superscript index n_m .

Suppose now that \mathbf{x}_{m_1} , \mathbf{x}_{m_2} , and \mathbf{x}_{m_3} denote three vertices connected to a vertex \mathbf{x}_m for every $m = 1, \dots, M$. Let γ_{mm_1} , γ_{mm_2} , and γ_{mm_3} be the grain boundary energy of the straight edges connecting \mathbf{x}_m with \mathbf{x}_{m_1} , \mathbf{x}_{m_2} , and \mathbf{x}_{m_3} , respectively. Then the law of the vertex motion (2.3) takes the form

$$(2.5) \quad \dot{\mathbf{x}}_m = \sum_{i=1}^3 \gamma_{mm_i} \frac{\mathbf{x}_{m_i} - \mathbf{x}_m}{\|\mathbf{x}_{m_i} - \mathbf{x}_m\|}, \quad m = 1, \dots, M.$$

In order to fully describe the evolution of the grain boundary network, it is still necessary to understand what happens during topological transitions that alter the structure of the network. This is the subject of the next section.

3. Topological transitions. As discussed earlier, a topological transition occurs when an element of a grain boundary network disappears. This can happen when the length of a single edge or area of a single grain decreases to zero. In this section, we formulate a set of rules that govern topological transitions. Our approach aims to improve on existing literature by emphasizing the consistency between continuous evolution of the network and discrete transitions.

Suppose first that an edge connecting two vertices \mathbf{x}_i and \mathbf{x}_j disappears at some time $t_0 > 0$. We will then say that \mathbf{x}_i and \mathbf{x}_j *collide* at time t_0 , forming a quadruple junction. When the grain boundary energy is weakly anisotropic, this junction is unstable in the following sense: it is possible to split the quadruple junction into two new triple junctions $\tilde{\mathbf{x}}_i$ and $\tilde{\mathbf{x}}_j$ connected by an infinitesimally short edge that will grow. Clearly, as indicated in Figure 1, the direction in which the splitting occurs is not the same as the direction of the original collision. We will refer to this event as a *neighbor switching*.

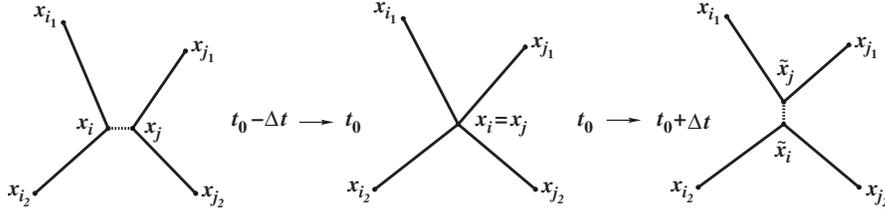


FIG. 1. Neighbor switching.

Next, we will use (2.5) to determine the orientation of the edge that forms as a result of neighbor switching. Fix a sufficiently small $\Delta t > 0$; then

$$(3.1) \quad \dot{\mathbf{x}}_i = \gamma_{ii_1} \frac{\mathbf{x}_{i_1} - \mathbf{x}_i}{\|\mathbf{x}_{i_1} - \mathbf{x}_i\|} + \gamma_{ii_2} \frac{\mathbf{x}_{i_2} - \mathbf{x}_i}{\|\mathbf{x}_{i_2} - \mathbf{x}_i\|} + \gamma_{ij} \frac{\mathbf{x}_j - \mathbf{x}_i}{\|\mathbf{x}_j - \mathbf{x}_i\|},$$

$$(3.2) \quad \dot{\mathbf{x}}_j = \gamma_{jj_1} \frac{\mathbf{x}_{j_1} - \mathbf{x}_j}{\|\mathbf{x}_{j_1} - \mathbf{x}_j\|} + \gamma_{jj_2} \frac{\mathbf{x}_{j_2} - \mathbf{x}_j}{\|\mathbf{x}_{j_2} - \mathbf{x}_j\|} + \gamma_{ij} \frac{\mathbf{x}_i - \mathbf{x}_j}{\|\mathbf{x}_i - \mathbf{x}_j\|},$$

when $t \in (t_0 - \Delta t, t_0)$. Note that the first two terms in both equations are continuous functions of t on $[t_0 - \Delta t, t_0]$ if we assume that \mathbf{x}_i and \mathbf{x}_j are continuous on $[t_0 - \Delta t, t_0]$. Subtracting (3.1) from (3.2) we have

$$(3.3) \quad (\mathbf{x}_j - \mathbf{x}_i)' = \mathbf{p}_-(t) - 2\gamma_{ij} \frac{\mathbf{x}_j - \mathbf{x}_i}{\|\mathbf{x}_j - \mathbf{x}_i\|},$$

where

$$(3.4) \quad \mathbf{p}_- = \gamma_{jj_1} \frac{\mathbf{x}_{j_1} - \mathbf{x}_j}{\|\mathbf{x}_{j_1} - \mathbf{x}_j\|} + \gamma_{jj_2} \frac{\mathbf{x}_{j_2} - \mathbf{x}_j}{\|\mathbf{x}_{j_2} - \mathbf{x}_j\|} - \gamma_{ii_1} \frac{\mathbf{x}_{i_1} - \mathbf{x}_i}{\|\mathbf{x}_{i_1} - \mathbf{x}_i\|} - \gamma_{ii_2} \frac{\mathbf{x}_{i_2} - \mathbf{x}_i}{\|\mathbf{x}_{i_2} - \mathbf{x}_i\|}$$

satisfies $\mathbf{p}_- \in C([t_0 - \Delta t, t_0])$. Let $\mathbf{x}_j - \mathbf{x}_i = \rho \mathbf{n}$, where $\rho = \|\mathbf{x}_j - \mathbf{x}_i\|$ and $\mathbf{n} = (\cos \theta, \sin \theta)$ and set $\boldsymbol{\tau} = (-\sin \theta, \cos \theta)$. Rewriting (3.3), we have

$$(3.5) \quad \dot{\rho} \mathbf{n} + \rho \dot{\theta} \boldsymbol{\tau} = \mathbf{p}_- - 2\gamma_{ij} \mathbf{n},$$

or by orthogonal decomposition,

$$(3.6a) \quad \dot{\rho} = \mathbf{p}_- \cdot \mathbf{n} - 2\gamma_{ij},$$

$$(3.6b) \quad \dot{\theta} = \frac{\mathbf{p}_- \cdot \boldsymbol{\tau}}{\rho}.$$

In Appendix 1, we use (3.6) to show that

$$\lim_{t \rightarrow t_0^-} \mathbf{p}_- \cdot \boldsymbol{\tau} = 0,$$

since $\lim_{t \rightarrow t_0^-} \rho = 0$. Thus $\mathbf{n}_- := \lim_{t \rightarrow t_0^-} \mathbf{n}$ must be parallel to $\mathbf{p}_-(t_0)$. The analogous arguments on $(t_0, t_0 + \Delta t)$ demonstrate that $\mathbf{n}_+ := \lim_{t \rightarrow t_0^+} \mathbf{n}$ must be parallel to $\mathbf{p}_+(t_0)$, where

$$(3.7) \quad \mathbf{p}_+ = \gamma_{jj_1} \frac{\mathbf{x}_{j_1} - \tilde{\mathbf{x}}_j}{\|\mathbf{x}_{j_1} - \tilde{\mathbf{x}}_j\|} + \gamma_{jj_2} \frac{\mathbf{x}_{j_2} - \tilde{\mathbf{x}}_j}{\|\mathbf{x}_{j_2} - \tilde{\mathbf{x}}_j\|} - \gamma_{ii_1} \frac{\mathbf{x}_{i_1} - \tilde{\mathbf{x}}_i}{\|\mathbf{x}_{i_1} - \tilde{\mathbf{x}}_i\|} - \gamma_{ii_2} \frac{\mathbf{x}_{i_2} - \tilde{\mathbf{x}}_i}{\|\mathbf{x}_{i_2} - \tilde{\mathbf{x}}_i\|},$$

per Figure 1. Note that the edge that had existed before the transition should disappear if the condition

$$(3.8) \quad \|\mathbf{p}_-(t_0)\| - 2\gamma_{ij} < 0$$

is satisfied. Further, an analogue of (3.6a) shows that

$$(3.9) \quad \|\mathbf{p}_+(t_0)\| - 2\gamma_{i\bar{j}} > 0$$

guarantees that the newly formed edge will grow.

Both of these inequalities simultaneously hold for collisions in grain boundary networks with isotropic grain boundary energy, and we expect them to hold in the case of weak anisotropy. Indeed, in grain growth simulations described below, we numerically observed that the condition (3.8) is always satisfied, as long as the anisotropy is not too strong. For large anisotropy quadruple junctions may become stable, as shown in section 8.6.

4. Collision time estimate. We can use the evolution equations (3.6) to estimate whether a pair of the adjacent vertices of the grain boundary network will collide during a given time step Δt . This estimate is essential to detect topological transitions within the numerical procedure that will be discussed in the subsequent sections.

Given the current time $t = t_c$, suppose that the edge connecting the vertices \mathbf{x}_i and \mathbf{x}_j becomes extinct at the time $t_c + t_{ext}$, where $t_{ext} < \Delta t$. Assuming that $\Delta t > 0$ is sufficiently small and using the continuity of \mathbf{p}_- on the interval $[t_c, t_c + t_{ext}]$, we have that $\mathbf{p}_-(t) = \mathbf{p}_-(t_c) + o(1)$. Now consider the system of equations

$$(4.1a) \quad \dot{\rho} = \mathbf{p} \cdot \bar{\mathbf{n}} - 2\gamma_{ij},$$

$$(4.1b) \quad \dot{\theta} = \frac{\mathbf{p} \cdot \bar{\boldsymbol{\tau}}}{\bar{\rho}}$$

on $[t_c, t_c + \bar{t}_{ext}]$ satisfying $\bar{\rho}(t_c) = \rho(t_c)$ and $\bar{\theta}(t_c) = \theta(t_c)$. Here $\mathbf{p} := \mathbf{p}_-(t_c)$ and $\bar{\rho}(\bar{t}_{ext}) = 0$. Taking the derivative of (4.1a), multiplying the resulting equation by $\bar{\rho}$, and using (4.1) we obtain

$$\begin{aligned} \bar{\rho}\ddot{\rho} &= \bar{\rho}\mathbf{p} \cdot \dot{\bar{\mathbf{n}}} = (\mathbf{p} \cdot \bar{\boldsymbol{\tau}})\bar{\rho}\dot{\theta} = (\mathbf{p} \cdot \bar{\boldsymbol{\tau}})^2 = \|\mathbf{p}\|^2 - (\mathbf{p} \cdot \bar{\mathbf{n}})^2 \\ &= \|\mathbf{p}\|^2 - (\dot{\rho} + 2\gamma_{ij})^2 = \|\mathbf{p}\|^2 - \dot{\rho}^2 - 4\gamma_{ij}\dot{\rho} - 4\gamma_{ij}^2. \end{aligned}$$

Rearranging terms then gives

$$(4.2) \quad (\bar{\rho}(\dot{\rho} + 4\gamma_{ij}))' = \|\mathbf{p}\|^2 - 4\gamma_{ij}^2.$$

Note that this equation no longer involves the angular coordinate $\bar{\theta}$.

Integrating (4.2) once leads to

$$(4.3) \quad \bar{\rho}(\dot{\rho} + 4\gamma_{ij}) = (\|\mathbf{p}\|^2 - 4\gamma_{ij}^2)(t - t_c) + \rho(t_c)(\dot{\rho}(t_c) + 4\gamma_{ij})$$

on $(t_c, t_c + \bar{t}_{ext})$. Suppose that $\|\mathbf{p}\|^2 - 4\gamma_{ij}^2 < 0$; then the right-hand side of (4.3) vanishes when $t - t_c = \rho(t_c)(\dot{\rho}(t_c) + 4\gamma_{ij}) / (4\gamma_{ij}^2 - \|\mathbf{p}\|^2)$. We claim that $\bar{\rho}$ becomes zero at the same time, i.e.,

$$\bar{t}_{ext} = \frac{\rho(t_c)(\dot{\rho}(t_c) + 4\gamma_{ij})}{4\gamma_{ij}^2 - \|\mathbf{p}\|^2} = \frac{\rho(t_c)(\mathbf{p}_-(t_c) \cdot \mathbf{n}(t_c) + 2\gamma_{ij})}{4\gamma_{ij}^2 - \|\mathbf{p}\|^2},$$

where the final expression follows from the definition of \mathbf{p} and (3.6a). Indeed, by our assumption that $\|\mathbf{p}\|^2 - 4\gamma_{ij}^2 < 0$ and from (4.1a), the expression $\dot{\bar{\rho}} + 4\gamma_{ij} = \mathbf{p} \cdot \bar{\mathbf{n}} + 2\gamma_{ij}$ is strictly positive and bounded on $(t_c, t_c + \bar{t}_{ext})$. It then follows that $\bar{\rho}$ vanishes along with the right-hand side of (4.3).

Finally, since $\mathbf{p}_-(t) = \mathbf{p}_-(t_c) + o(1)$ on $[t_c, t_c + t_{ext}]$, we have that \bar{t}_{ext} is the leading order approximation to t_{ext} , i.e.,

$$(4.4) \quad t_{ext} = \frac{\rho(t_c) (\mathbf{p}_-(t_c) \cdot \mathbf{n}(t_c) + 2\gamma_{ij})}{4\gamma_{ij}^2 - \|\mathbf{p}_-(t_c)\|^2} (1 + o(1)).$$

5. Stability analysis. Here we present arguments to show that the explicit numerical scheme proposed in this paper is stable. For simplicity, consider a discretization of the system (3.6) in the isotropic case when the grain boundary energy is identically equal to one:

$$(5.1) \quad \begin{cases} \rho_{i+1} = \rho_i + (\mathbf{p}_-(t_i) \cdot \mathbf{n}(t_i) - 2) \Delta t, \\ \theta_{i+1} = \theta_i + \frac{\Delta t}{\rho_i} \mathbf{p}_-(t_i) \cdot \boldsymbol{\tau}(t_i). \end{cases}$$

To simplify this system further, suppose that $\mathbf{p}_-(t_i) = \mathbf{p} = p(\cos \theta_p, \sin \theta_p) = \text{const}$ for all $i = 1, 2, 3, \dots$ and $|\theta_0 - \theta_p| \ll 1$. Then the system (5.1) takes the form

$$(5.2) \quad \begin{cases} \rho_{i+1} = \rho_i + (p \cos(\theta_p - \theta_i) - 2) \Delta t, \\ \theta_{i+1} = \theta_i + \frac{p\Delta t}{\rho_i} \sin(\theta_p - \theta_i). \end{cases}$$

Linearization of the system (5.2) in θ_i around θ_p gives

$$(5.3) \quad \begin{cases} \rho_{i+1} = \rho_i + (p - 2)\Delta t, \\ \theta_{i+1} - \theta_p = \left(1 - \frac{p}{\rho_i} \Delta t\right) (\theta_i - \theta_p). \end{cases}$$

If $p < 2$, then ρ should vanish after $N := \lfloor \rho_0(2 - p)^{-1} \Delta t^{-1} \rfloor$ time steps. Possible problems with stability may, therefore, arise when N is large, that is, when p is close to 2. This situation corresponds to a local equilibrium of the grain boundary network when all angles between adjacent edges are close to 120° in the vicinity of the disappearing edge. Consider the worst-case scenario when p remains close to 2 for a long time (this is unlikely in real simulations as the motion of other vortices will likely cause p to change). Suppose that Δt satisfies the condition $0 < p \Delta t / \rho_0 < 1$; then

$$\Delta t = \frac{\alpha \rho_0}{p},$$

where $0 < \alpha < 1$. Substituting this expression into (5.3), the second equation in (5.3) takes the form

$$(5.4) \quad \theta_{i+1} - \theta_p = \left(1 - \frac{\alpha}{1 - \frac{i}{x}}\right) (\theta_i - \theta_p),$$

where $x = \frac{p}{\alpha(2-p)}$. Using the same notation, we have $N = \lfloor x \rfloor$. It now follows that

$$(5.5) \quad \theta_i - \theta_p = (\theta_0 - \theta_p) \prod_{j=1}^i \left(1 - \frac{\alpha}{1 - \frac{j}{x}}\right),$$

where $i = 1, 2, 3, \dots, \lfloor x \rfloor - 1$ since we do not need to determine θ when $\rho = 0$. Suppose that $p \uparrow 2$; then $x \rightarrow \infty$ and the magnitude of the factors in the product in (5.5) is close to $1 - \alpha$ when j is small. On the other hand, when j is close to $\lfloor x \rfloor$,

$$\left| 1 - \frac{\alpha}{1 - \frac{j}{x}} \right| \gg 1,$$

and the value of the product is largest when $i = \lfloor x \rfloor - 1$. Thus there are no issues with numerical stability if the product

$$(5.6) \quad \Phi(\alpha, x) := \prod_{j=1}^{\lfloor x \rfloor - 1} \left| 1 - \frac{\alpha}{1 - \frac{j}{x}} \right|$$

remains finite for large x . If $\Gamma(x)$ is the Γ -function, in Appendix 2 we show that

$$(5.7) \quad \Phi(\alpha, x) = \frac{1}{\pi} \frac{\Gamma(1 + x - \lfloor x \rfloor) \Gamma((1 - \alpha)x) \Gamma(\lfloor x \rfloor - x + \alpha x)}{\Gamma(x)} \sin \pi ((1 - \alpha)x - \lfloor (1 - \alpha)x \rfloor),$$

as long as $\alpha x > 1$. Since $\max_{1 \leq \lambda \leq 2} \Gamma(\lambda) = 1$, we have that

$$0 \leq \Phi(\alpha, x) \leq \frac{1}{\pi} \frac{\Gamma((1 - \alpha)x) \Gamma(\lfloor x \rfloor - x + \alpha x)}{\Gamma(x)},$$

for all $x > 0$ as long as $\alpha x > 1$. Further, if $0 < \alpha < 1$ is fixed and $x \gg 1$, then the monotonicity and asymptotics of Γ for large values of its argument imply that

$$0 \leq \Phi(\alpha, x) \leq \frac{1}{\pi} \frac{\Gamma((1 - \alpha)x) \Gamma(\alpha x)}{\Gamma(x)} \sim \left(\alpha^\alpha (1 - \alpha)^{1 - \alpha} \right)^x \rightarrow 0 \text{ as } x \rightarrow 0.$$

Then $\theta_{N-1} \rightarrow 0$ as $p \rightarrow 2$ and the numerical scheme is stable.

6. Vertex code algorithm description. The main algorithm can be decomposed into two parts: discrete and continuous, describing topological transitions and motion of triple junctions, respectively. Both of these processes depend on the time resolution Δt . The discrete component of the algorithm detects and carries out topological transitions within $[t, t + \Delta t]$, while the continuous component evolves the triple junctions from time t to time $t + \Delta t$. The procedure is described in detail in Algorithm 1 below.

The continuous part of the evolution is relatively straightforward and can be achieved by solving a system of ODEs by means of any available numerical scheme, e.g., MATLAB `ode45` routine. The principal aim of this implementation is to isolate and fully resolve topological transitions. This is done by dynamically adapting the time step Δt using the formula (4.4) within the following procedure.

Initially, Δt is set equal to a prescribed value of Δt_0 . On each time step, the algorithm estimates extinction times for all grain boundaries and selects only those which fall in the interval $[t, t + \Delta t]$. Then the corresponding grain boundaries are sorted according to their extinction times. Next we move along this list and record those boundaries whose vertices have not yet been encountered. This process continues until either all vertices are exhausted or a boundary with an already recorded vertex has been detected. In the latter case, the process stops and the time step is adjusted to only allow extinction of the boundaries that have been recorded. This ensures that

ALGORITHM 1. MAIN ALGORITHM.

```

1: GRAINS(0,:) ← Initial configuration at time  $t = 0$ .
2:  $\Delta t_0 \leftarrow$  Upper bound for  $\Delta t$ .
3:  $\Delta t \leftarrow \Delta t_0$ 
4: while Stopping Criteria Not Satisfied do
5:    $t_{ext} \leftarrow$  Compute extinction times for all grain boundaries.
6:    $\mathcal{L}_1 \leftarrow$  Select grain boundaries such that  $0 < t_{ext} < \Delta t$ 
7:   if  $\mathcal{L}_1$  is empty then
8:     GRAINS( $t + \Delta t$ ,:) ← Evolve grain data structure GRAINS( $t$ ,:) to time
      $t + \Delta t$ .
9:   else
10:     $\mathcal{L}_1 \leftarrow$  Sort  $\mathcal{L}_1$  in increasing order of  $t_{ext}$ 
11:    tmp ← {},  $\mathcal{L}_2 \leftarrow \{\}$ ,  $l_2 \leftarrow \{\}$ 
12:    for  $l \in \mathcal{L}_1$  do
13:      if triple junctions( $l$ )  $\cap$  tmp =  $\emptyset$  then
14:         $\mathcal{L}_2 \leftarrow$  Add grain boundary  $l$ .
15:        tmp ← Add triple junctions( $l$ ).
16:         $l_2 \leftarrow l$ .
17:      else
18:         $\Delta t \leftarrow \frac{t_{ext}(l_2) + t_{ext}(l)}{2}$ .
19:        Break-loop.
20:      end if
21:    end for
22:    for  $m \in \mathcal{L}_2$  do
23:       $nl, nr \leftarrow$  Numbers of sides of the grains adjacent to the grain boundary
       $m$ .
24:      if  $nl = 3$  and  $nr = 3$  then
25:        return ERROR
26:      else if  $nr = 3$  or  $nl = 3$  then
27:        Remove the adjacent 3-sided grain. Replace it with an edge, as
        shown in Figure 3
28:      else
29:        Flip the grain boundary  $m$ , as shown in Figure 2.
30:      end if
31:      GRAINS( $t + \Delta t, m$ ) ← Evolve GRAINS( $t, m$ ) to time  $t + \Delta t$ .
32:    end for
33:    GRAINS( $t + \Delta t, \neg\mathcal{L}_2$ ) ← Evolve GRAINS( $t, \neg\mathcal{L}_2$ ) to time  $t + \Delta t$ .
34:     $\Delta t \leftarrow \Delta t_0$ .
35:  end if
36: end while

```

there is spatial separation between topological transitions. Once the time step has been adapted, transitions that involve all recorded boundaries are implemented and the rest of the network is allowed to evolve in a continuous fashion. Then the time step is reset to the original value of Δt_0 and the procedure is repeated until 80% of the grains have been eliminated.

There are two types of transition events: (1) neighbor switching and (2) grain removal. From the detection point of view outlined above, these events are

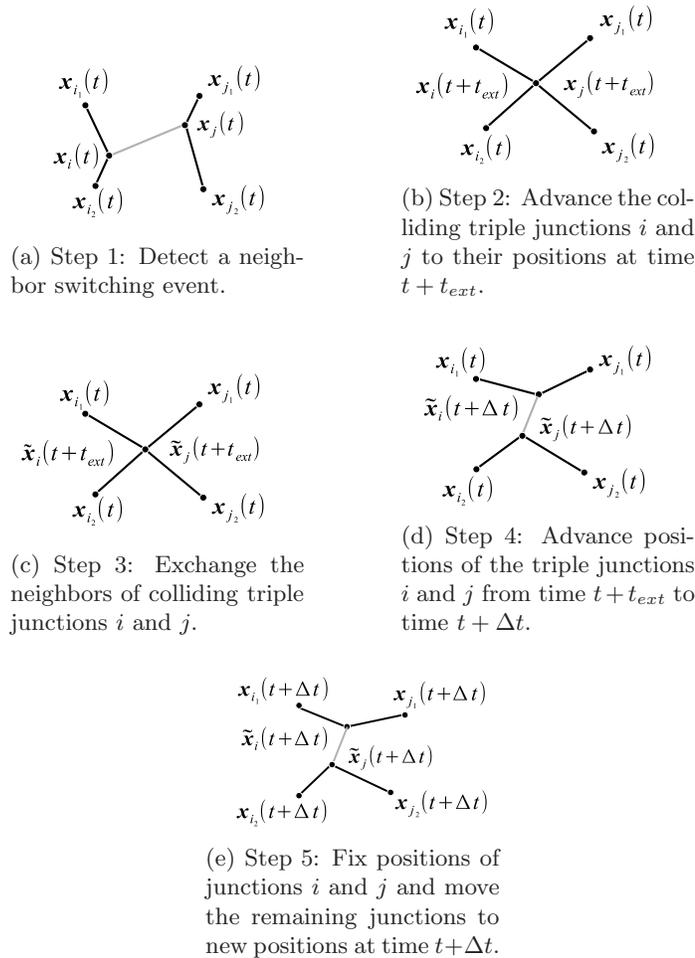
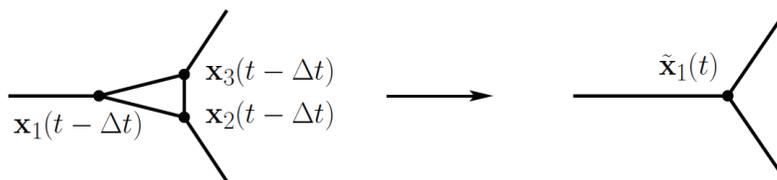


FIG. 2. Neighbor switching algorithm.

FIG. 3. Configurations resulting in grain removal. The location of the new triple junction $\tilde{x}_1(t)$ is set to be the collision point for the two triple junctions that collide first.

indistinguishable. On the other hand, the ways in which these events are resolved are completely different. At first, all detected events are considered to be a neighbor switching event, as shown in Figure 2. However, when one of the two grains adjacent to the disappearing grain boundary has only three sides, we proceed to remove that grain, as depicted in Figure 3. The neighbor switching is done according to the

rules discussed in section 3. We assume that the anisotropy is small enough so that a shrinking grain boundary cannot be simultaneously adjacent to two 3-sided grains, and the procedure returns an error if this happens.

Note that only 3-sided grains are allowed to be removed. Thus, for a 5-sided grain to be able to disappear, it needs first to become a 4-sided grain and then a 3-sided grain. The number of sides of a given grain can change either through a neighbor switching event or as a consequence of a neighboring grain disappearance. Finally, when a switching of neighbors is performed, the length of the new grain boundary is computed to be proportional to $\Delta t - t_{ext}$, where t_{ext} is the extinction time of the grain boundary in question.

7. Numerical convergence study. Although a rigorous analytical investigation is beyond the scope of this paper, in this section we present a numerical study subjecting the proposed algorithm to several tests for accuracy and convergence. The first test is designed to check how well the numerical procedure is able to handle topological transitions via time step adaptation. For this purpose, we compare the results of several simulations using different values of the maximum time step Δt_0 in Figures 4 and 5. Figure 4 shows grain boundary structures obtained from the same initial configuration with 200 grains at $t = 0.45$ when approximately 80% of the grains were removed. Three different simulations with maximum time step sizes $\Delta t_0 = 10^{-2}$, 10^{-5} , and 10^{-6} were performed. It is evident that, with the exception of the grain structure obtained when $\Delta t_0 = 10^{-2}$ —the coarsest maximum time step—all grain boundaries lie directly on top of each other. In Figure 5 we plot the dependence between the simulation time and the size of time step for $\Delta t_0 = 10^{-2}$ and 10^{-6} . The figure shows that fewer time step refinements are performed for smaller Δt_0 and that for a very small Δt_0 , refinements are needed only when a grain removal occurs.

Next we test our algorithm for convergence using the following measures. First we run a set of simulations corresponding to several Δt_0 . We consider the simulation with the smallest $\Delta t_0 = 10^{-6}$ as “well-resolved” and benchmark the results of other simulations against it. In Figure 6 we consider the difference between the positions of triple junctions produced by a given and the well-resolved simulations at the time $t = 0.45$. Only the remaining common junctions are considered in this calculation. It can be seen that the error decreases linearly toward zero as $\Delta t_0 \rightarrow 0$. In fact, the number of noncommon triple junctions remaining at time t decreases to zero as well and it becomes exactly zero when $\Delta t_0 = 10^{-4}$ (Figure 7).

These tests allow the conclusion that the proposed algorithm successfully handles both topological transitions and the grain boundary motion and is numerically stable.

8. Numerical results.

8.1. Statistics. In this section we analyze the statistics for networks with isotropic and weakly anisotropic grain boundary energy. In all cases the network is initialized via a Voronoi tessellation using points uniformly distributed in the computational domain and assuming periodic boundary conditions. All networks initially contained 100,000 grains and were evolved until 80% of the grains were removed.

Figure 8 depicts the relative area distribution using linear and log scales. We observe that the distribution is skewed toward grains with smaller areas (this is emphasized in the log-scale plot). There is a notable difference from statistics produced by a curvature-driven simulation; in fact, grains with smaller areas tend to have a smaller rate of area change (Figure 18(a)). As a result, a grain boundary network evolved via triple junctions motion tends to have more small grains.

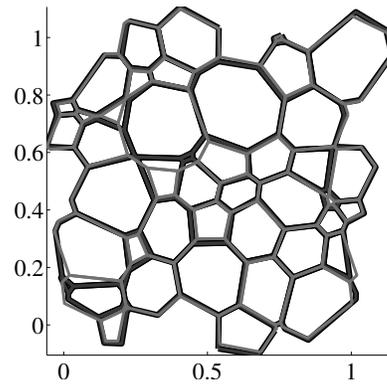


FIG. 4. Grain boundary networks evolved from the same configuration with 200 grains up to $t = 0.45$ when approximately 80% of the grains were removed. The simulations were run with three different maximum time steps: $\Delta t_0 = 10^{-2}$ (light gray), 10^{-5} (dark gray), and 10^{-6} (black).

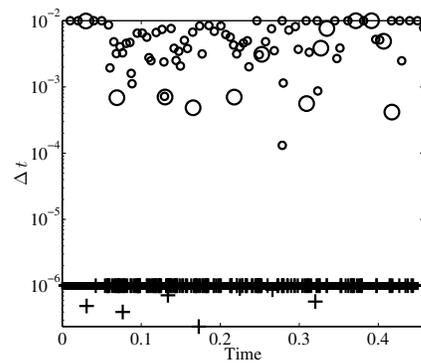


FIG. 5. Evolution of the time step for $\Delta t_0 = 10^{-2}$ (\circ) and $\Delta t_0 = 10^{-7}$ ($+$). Larger markers represent the instances when the decrease in time step was caused by grain removal in the absence of neighbor switching.

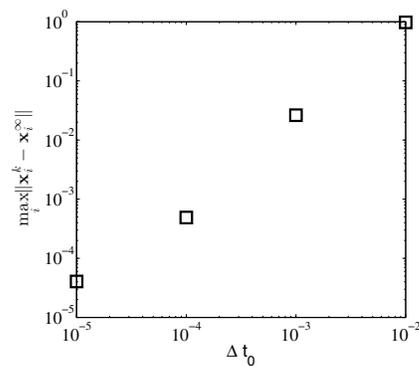


FIG. 6. Deviation between positions of triple junctions corresponding to grain boundary networks for various Δt_0 compared to a well-resolved simulation with $\Delta t_0 = 10^{-7}$. All simulations started from the same initial data and were compared at the same absolute time $t = 0.45$.

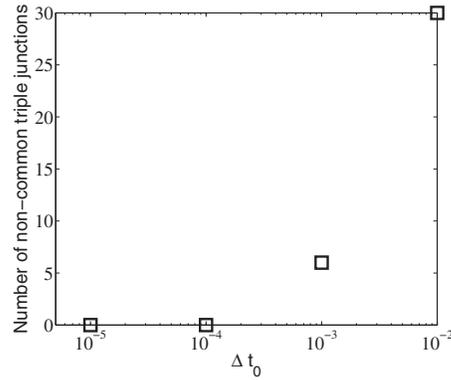


FIG. 7. Number of noncommon triple junctions between grain boundary networks for various Δt_0 and a well-resolved simulation with $\Delta t_0 = 10^{-6}$. All simulations started from the same initial data and were compared at the same absolute time $t = 0.45$.

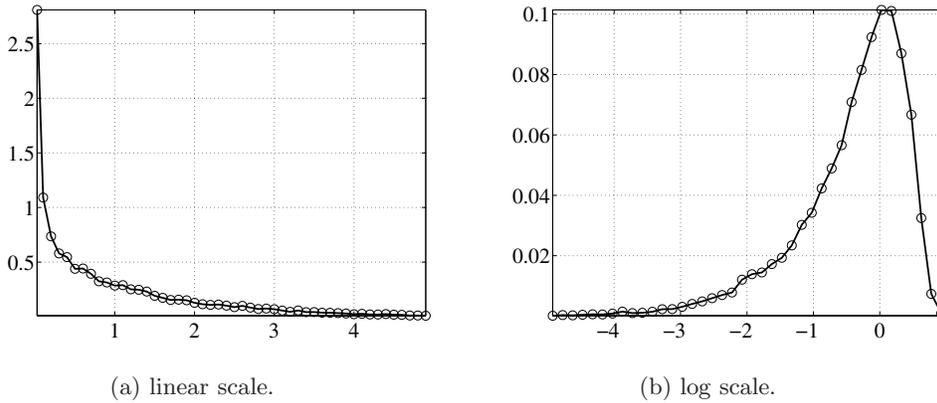


FIG. 8. Relative grain area distribution in the isotropic network consisting of 20,000 grains that evolved from an initial configuration of 100,000 grains.

The distribution of the number of sides per grain is shown in Figure 9(a) and demonstrates a bias toward 3-sided grains. We attribute this to the fact that grains need to go through a cascade of decreasing number of sides before they can be removed, i.e., a grain needs to become a 3-sided grain before it is allowed to disappear. The largest proportion of grains are 5-sided. Another important feature is that when the number of sides exceeds $n = 12$, the value of the probability density function is very small compared to the values when $n \leq 12$. This is relevant to understanding statistics presented below because the sample size for large n is small.

The third statistic (Figure 9(b)) is the dihedral angle distribution. This distribution does not seem to be centered at 120° but instead shows a small shift toward angles larger than 120° .

Finally, we computed the average number of sides of neighbors for grains with a given number of sides (Figure 10(a)) as well as the reduced average area of the neighbors (Figure 10(b)). A comparison to the empirical Aboav and Aboav–Weaire laws [28] is shown in Figure 11 for the systems from which the 20% and 80% of the grains were removed, respectively. The Aboav–Weaire law appears to provide a better fit in both cases.

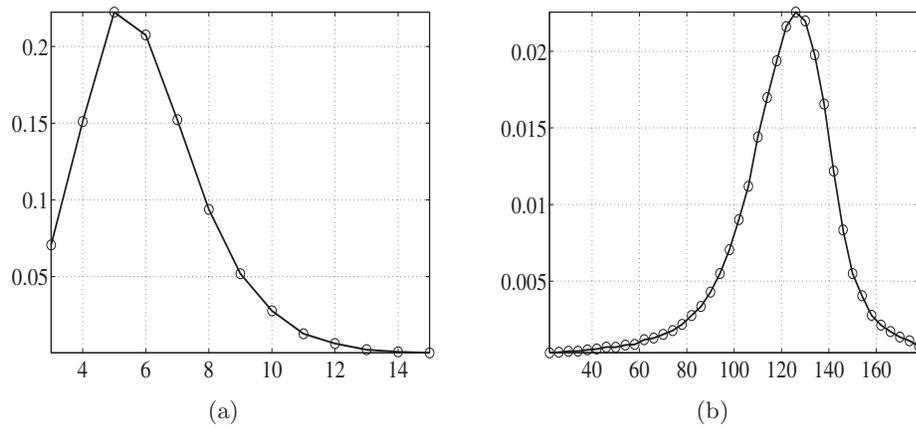


FIG. 9. Distributions for the (a) number of sides of a grain and (b) dihedral angle. The data is collected using the isotropic network of 20,000 grains that evolved from an initial configuration consisting of 100,000 grains.

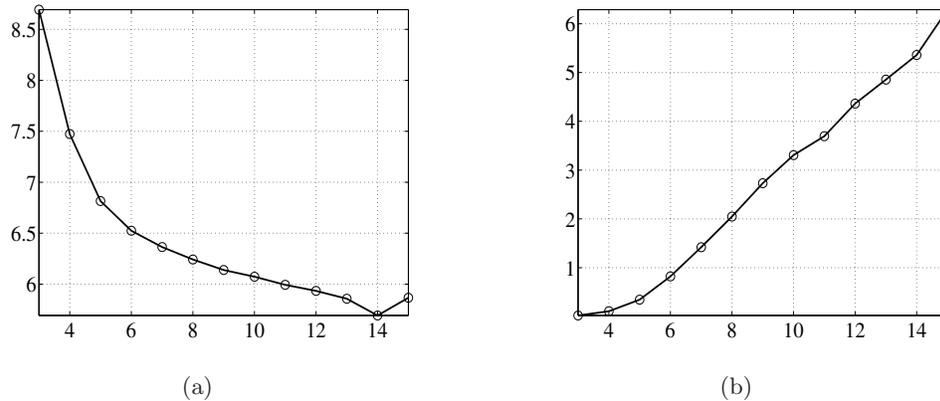


FIG. 10. For grains with a given number of sides: (a) average number of sides of neighboring grains; (b) average relative area of neighboring grains. The data is collected using the isotropic network of 20,000 grains that evolved from an initial configuration consisting of 100,000 grains.

The distributions obtained for grain boundary networks with weakly anisotropic grain boundary energy $\gamma(\Delta\alpha) = 0.95 - 0.05 \cos^3(4\Delta\alpha)$ are shown in Figures 12–14. Note that the statistical features that develop in this case are essentially identical to those observed when the grain boundary energy is isotropic.

8.2. Self-similarity. We have tested the distributions we have observed for self-similarity. We found that self-similarity does indeed develop and is very consistent for all distributions of interest. Here a critical issue is that a sufficiently large number of grains is needed initially for a clear trend to develop. In our experience, the statistics do not depend on the details of the initial configuration; these can only affect the time needed to achieve stable statistics. Here the initial set of grains was generated using the Voronoi construction for a uniformly distributed random collection of points in a rectangular domain.

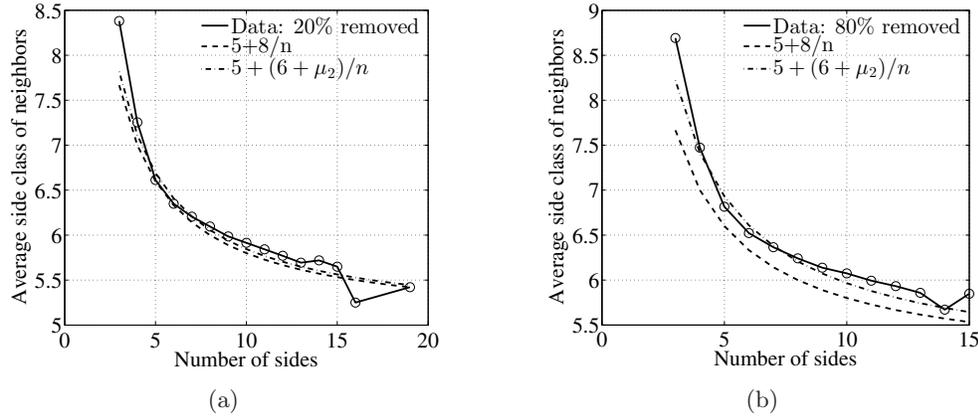


FIG. 11. Comparison with Aboav and Aboav-Weaire laws for an isotropic network: (a) 20% of the grains have been removed; (b) 80% of the grains have been removed.

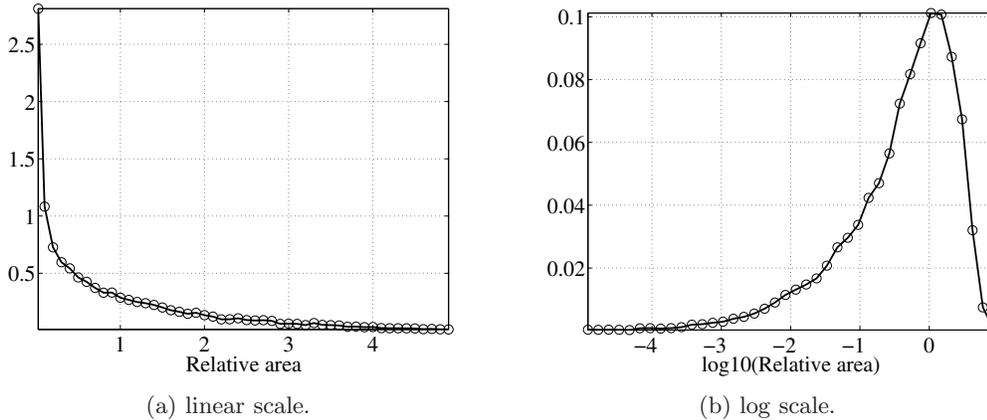


FIG. 12. Relative grain area distribution in the anisotropic network consisting of 20,000 grains that evolved from an initial configuration of 100,000 grains.

Figures 15 and 16 show the development of various distributions over time. The plots were produced by evolving grain boundary networks that initially contained 100,000 grains up until 20,000 grains remained.

8.3. Comparison of different neighbor switching rules. We have compared three types of neighbor switching rules used in the literature: maximum dissipation rate [7], maximum force [10], and the proposed approach. Figure 17 shows a part of the grain boundary network immediately preceding the first neighbor switching event and for several time steps afterward. The three rules result in a different initial orientation of the newly formed grain boundary that has length proportional to $\Delta t - t_{ext}$. In all cases, subsequent evolution of the network corrects the angle to the one enforced by the continuous part of the algorithm. Overall, it appears that a “suboptimal” neighbor switching rule leads to accumulating errors that result in a grain boundary network that differs significantly from that produced using the “optimal” rule, starting from the same initial conditions. However, singular behavior of the continuous part of the dynamical system at the time of the neighbor switch, combined with a smaller

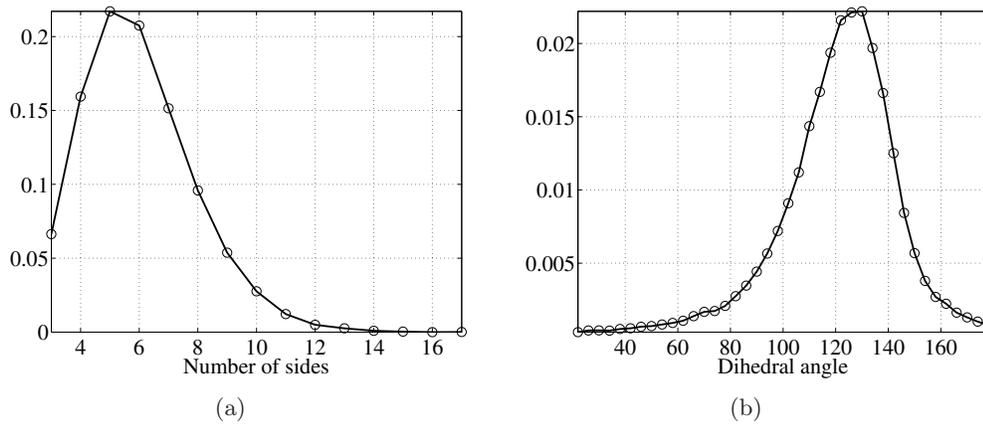


FIG. 13. Distributions for the (a) number of sides of a grain and (b) dihedral angle. The data is collected using the anisotropic network of 20,000 grains that evolved from an initial configuration consisting of 100,000 grains.

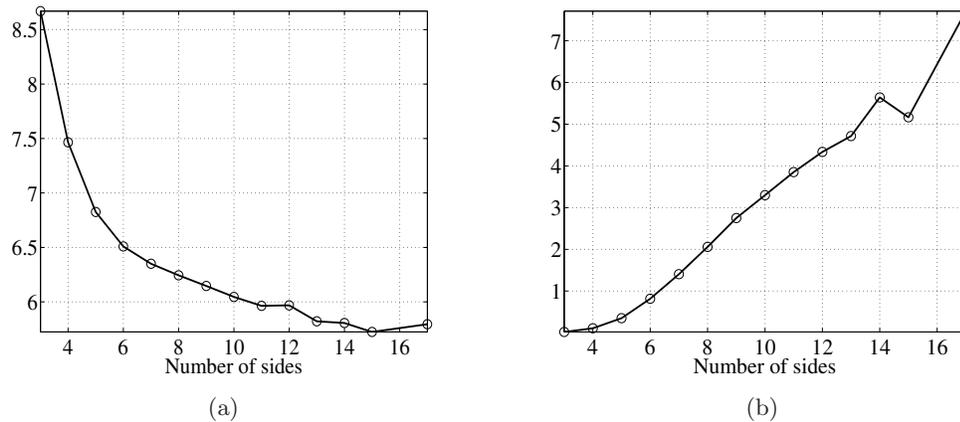


FIG. 14. For grains with a given number of sides: (a) average number of sides of neighboring grains; (b) average relative area of neighboring grains. The data is collected using the anisotropic network of 20,000 grains that evolved from an initial configuration consisting of 100,000 grains.

maximum time step Δt_0 , makes this error less significant, per Figure 17(d). In all cases, the statistical features of the network seem to be unaffected by the type of the rule used.

From our simulations, it also appears that the networks evolving via different neighbor switching rules (or via the same rule, but with a different maximum time step) move through a very similar sequence of configurations in the state space, albeit at different times.

8.4. Rate of area change for an N -sided grain. A well-known result for curvature-driven grain growth is the von Neumann–Mullins $(n - 6)$ -rule. The rule states that, given constant mobility and constant anisotropy, and assuming that the network satisfies the Herring condition (angles between the boundaries meeting at a triple junction are all equal to 120° in an isotropic case), we have

$$\frac{dA}{dt} = c(n - 6),$$

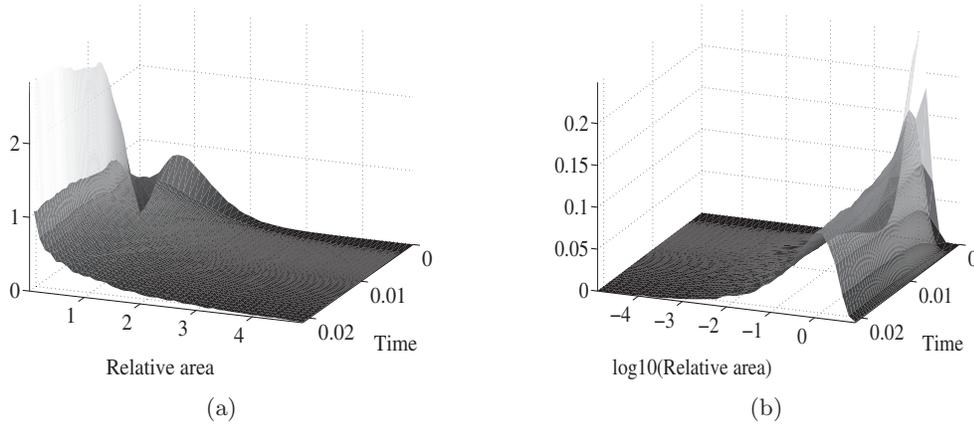


FIG. 15. Evolution in time of the relative grain area distribution. The statistics were collected for every time step while evolving grain boundary networks that initially contained 100,000 grains up until 20,000 grains remained.

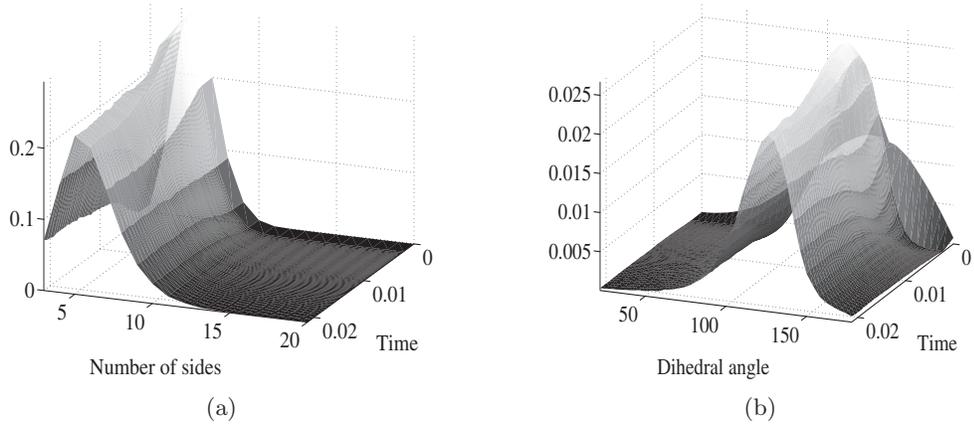


FIG. 16. Evolution in time of distributions for the (a) number of sides of a grain and (b) dihedral angle. The statistics were collected for every time step while evolving grain boundary networks that initially contained 100,000 grains up until 20,000 grains remained.

where $c > 0$ is a known constant. In a vertex algorithm the $n - 6$ rule does not hold. Indeed, in Figure 18(a) we observe that, although the relation for n between 5 and 15 is close to being linear, it is far from that for grains with a smaller number of sides. The distribution depends on time and has a self-similar shape; however, the nature of the observed dependence is still an open problem that will be addressed in a future publication. Figure 18(b) shows that the average area for each class of grains with a given number of sides grows over time. This same behavior is observed in curvature codes [7].

8.5. Stability. We have tested for stability distributions that develop for networks evolving via our algorithm in both isotropic and anisotropic cases. Here we only present the results for the isotropic case due to space constraints and because the conclusions are qualitatively similar.

The test has been performed on a sample with 100,000 grains initially, where the simulation was run until 50% of grains were removed. We decomposed the resulting

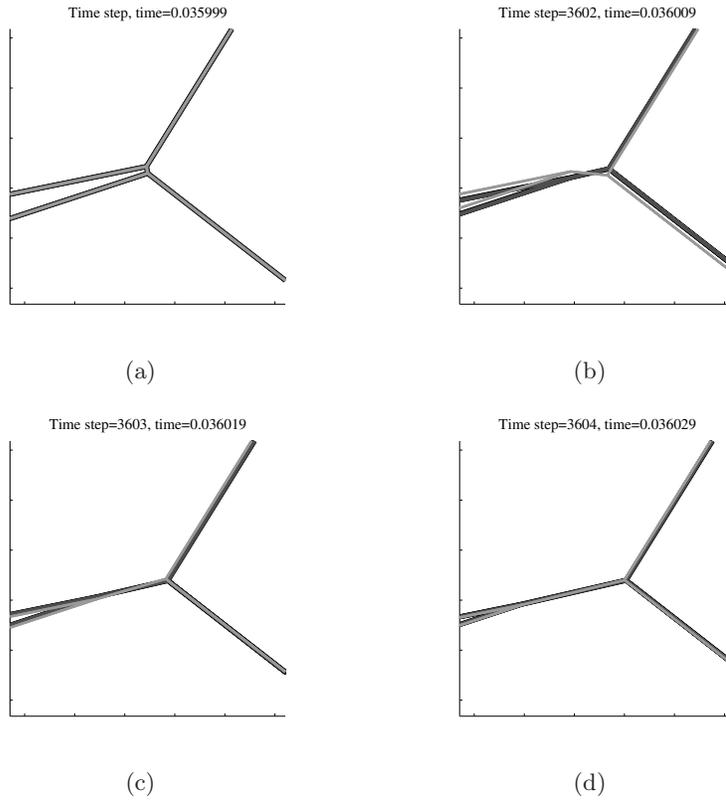


FIG. 17. Comparison of three neighbor switching rules: Maximum dissipation rate (light gray) [7], maximum force (gray) [10], and the approach described in this paper (black). After a short period of time, configurations are essentially indistinguishable when using a small time step.

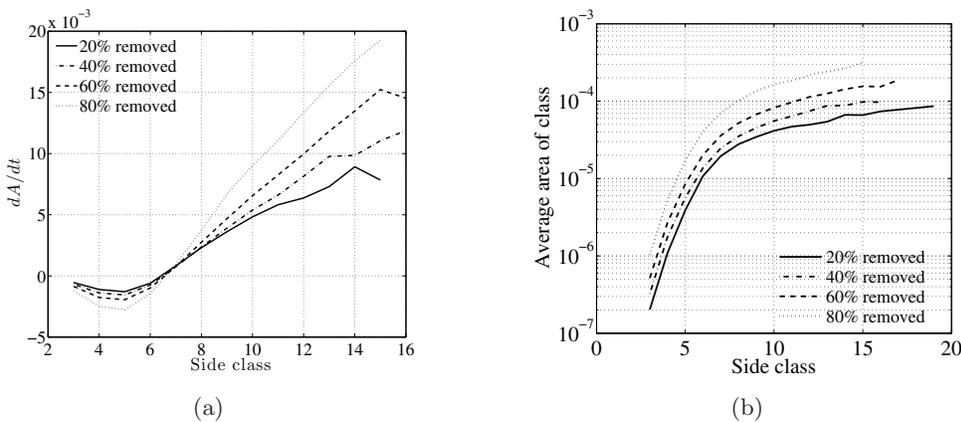


FIG. 18. (a) Average rate of change of grain area versus the number of sides of the grain. (b) Average grain area versus the number of sides of the grain.

sample into 12 spatially smaller subsets of equal area and collected various statistics for each sample. These were compared to the output of 10 simulations with initially 20,000 grains that were run until only 4,000 were left. The statistics were computed for these smaller samples. We also analyzed the statistics for a single 50,000 sample.

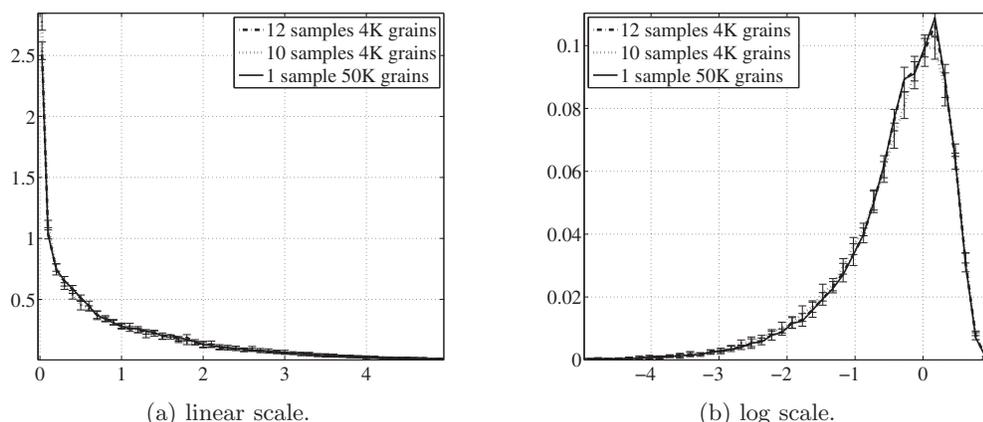


FIG. 19. Stability of the relative grain area distribution. The average of distributions for 12 subsets of the same simulation, the average of distributions for 10 different simulations, and the distribution for one large simulation are shown. The deviation of distributions from their average for each group is indicated by error bars.

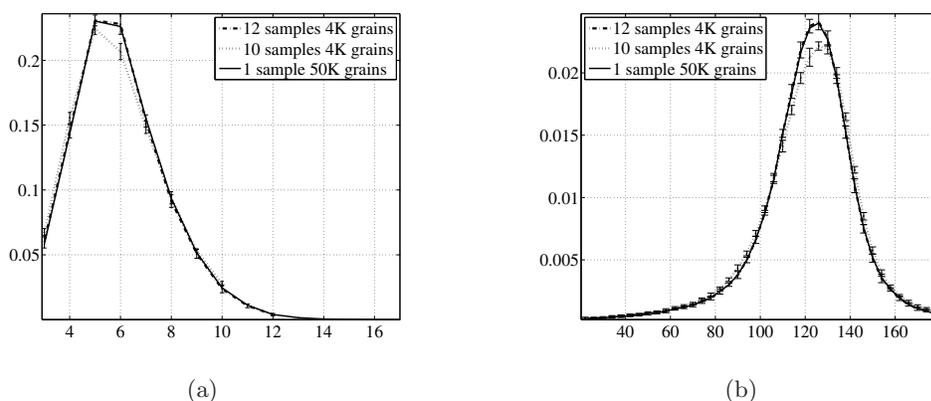


FIG. 20. Stability of distributions for the (a) number of sides of a grain and (b) dihedral angle. The average of distributions for 12 subsets of the same simulation, the average of distributions for 10 different simulations, and the distribution for one large simulation are shown. The deviation of distributions from their average for each group is indicated by error bars.

Figures 19 through 21 present the outcome of this study. This experiment shows numerical evidence that all distributions are remarkably stable in the sense that collecting statistics over subareas of the network or the entire network produces the same results.

8.6. Quadruple junctions and their stability. In a network with a large anisotropy stable quadruple junctions may exist. From the implementation point of view, they are two triple junctions that almost overlap. In principle, one needs to develop a separate set of rules that govern neighbor switching for this type of a junction. In our algorithm, quadruple junctions are always assumed to split into triple junctions. However, if a quadruple junction is stable, then any new boundary created as a result of the split would shrink and disappear, restoring the original quadruple junction. In this way, the code is capable of dealing with stable quadruple junctions. The example of such junction in a grain boundary network with an anisotropic grain boundary energy $\gamma(\Delta\alpha) = 0.55 - 0.45 \cos^3(4\Delta\alpha)$ is shown in Figure 22.

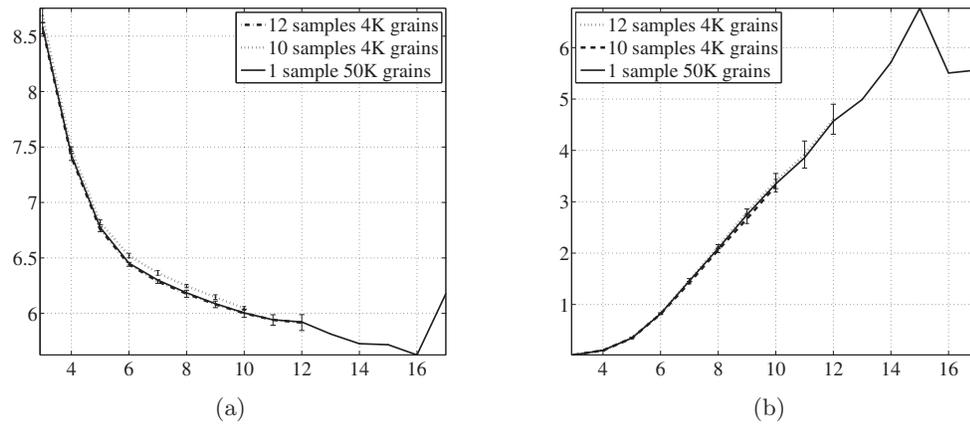


FIG. 21. For the grains with a given number of sides: (a) average number of sides of neighbors; (b) average relative grain area of neighbors. The average of distributions for 12 subsets of the same simulation, the average of distributions for 10 different simulations, and the distribution for one large simulation are shown. The deviation of distributions from their average for each group is indicated by error bars.

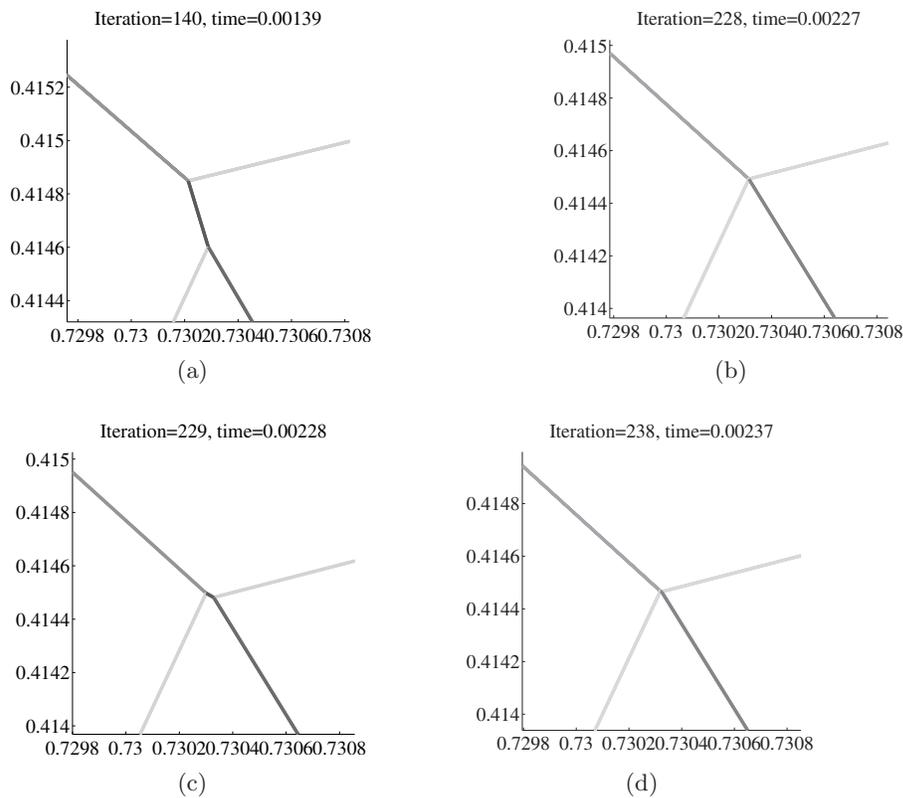


FIG. 22. Evolution of two triple junctions that remain together and emulate a quadruple junction. Darker colors correspond to larger grain boundary energy.

9. Conclusions. We have developed a numerical algorithm for an evolving grain boundary network described by a simple vertex model. The model can be formally derived via the assumption that the mobility of triple junctions is much lower than the mobility of grain boundaries evolving via curvature-driven motion. We have used a semirigorous analysis of vertex dynamics to derive the neighbor switching rules consistent with continuous evolution of vertices as well as the estimates of vertex collision times. These estimates were incorporated into the numerical procedure to pinpoint the times corresponding to topological transitions. By simulating coarsening of the vertex model grain boundary network, we have demonstrated that the geometry of configurations that develop is described by the standard statistical measures for characterizing grain growth. These include distributions of relative areas of grains, dihedral angle, and number of sides, among others. We have confirmed spatiotemporal stability of the distributions that develop in a network evolving via our numerical algorithm. We find that the distributions are essentially independent of the level of numerical resolution as the network passes through the sequence of similar states. While mesoscopic characteristics of the network appear to be robust, even with respect to changes in the rules governing topological transitions, the microscopic features of the network at a given time are sensitive to any modifications of the algorithm.

Appendix 1. Suppose that the edge connecting two vertices \mathbf{x}_i and \mathbf{x}_j disappears at the time t_0 and that \mathbf{p}_- , ρ , θ , \mathbf{n} , and $\boldsymbol{\tau}$ are as defined in section 3. Here we will use (3.6) to argue that, as long as $\mathbf{p}_- \in C([t_0 - \Delta t, t_0])$ for some small $\Delta t > 0$ and

$$\|\mathbf{p}_-(t_0)\| - 2\gamma_{ij} < 0,$$

we have that

$$(9.1) \quad \lim_{t \rightarrow t_0^-} \mathbf{p}_- \cdot \boldsymbol{\tau} = 0.$$

First, since $\mathbf{p}_- \in C([t_0 - \Delta t, t_0])$ and $\|\mathbf{p}_-(t_0)\| - 2\gamma_{ij} < 0$, we can choose Δt small enough so that $-4\gamma_{ij} \leq \mathbf{p}_- \cdot \mathbf{n} - 2\gamma_{ij} \leq -\alpha$ on $[t_0 - \Delta t, t_0]$ for some $\alpha > 0$. The system (3.6) has a continuous solution on $[t_0 - \Delta t, t_0]$, where the function ρ is, in fact, continuous on $[t_0 - \Delta t, t_0]$. Integrating (3.6a) and using the condition $\rho(t_0) = 0$, we obtain

$$(9.2) \quad \alpha(t_0 - t) \leq \rho(t) \leq 4\gamma_{ij}(t_0 - t),$$

when $t \in [t_0 - \Delta t, t_0]$.

Set $\mathbf{p}_- = \|\mathbf{p}_-\|(\cos \theta_p, \sin \theta_p)$; then (3.6b) takes the form

$$(9.3) \quad \dot{\theta} = \frac{\|\mathbf{p}_-\|}{\rho} \sin(\theta_p - \theta).$$

We will assume here that $\mathbf{p}_- \neq \mathbf{0}$ on $[t_0 - \Delta t, t_0]$; then both $\|\mathbf{p}_-\|$ and θ_p are continuous and $\|\mathbf{p}_-(t)\| > \beta$ for all $t \in [t_0 - \Delta t, t_0]$ and some constant $\beta > 0$.

Suppose first that $0 \leq \theta_p - \theta \leq \pi$ on $[t_0 - \Delta t, t_0]$. Then $\dot{\theta} \geq 0$ on $t \in [t_0 - \Delta t, t_0]$, hence $\lim_{t \rightarrow t_0^-} \theta$ exists. If this limit is finite, it immediately follows from integrability of the right-hand side of (9.3) and (9.2) that the $\lim_{t \rightarrow t_0^-} \theta = \theta_p(t_0)$. If the $\lim_{t \rightarrow t_0^-} \theta = \infty$, then the assumption that $\theta_p - \theta > 0$ implies that $\lim_{t \rightarrow t_0^-} \theta_p = \infty$; this violates the continuity of θ_p on $[t_0 - \Delta t, t_0]$. An analogous argument can be used to show a symmetric result for $-\pi \leq \theta_p - \theta \leq 0$.

It remains to prove (9.1) when there exists a sequence $t_n \rightarrow t_0$ such that $t_n < t_{n+1}$ and $\sin(\theta_p(t_n) - \theta(t_n)) = 0$ for all $n = 1, 2, 3, \dots$. Since θ_p is continuous on $[t_0 - \Delta t, t_0]$, we can assume that

$$(9.4) \quad |\theta_p(t_n) - \theta_p(t_0)| < \pi/2$$

for all $n = 1, 2, 3, \dots$. Fix an arbitrary $n = 1, 2, 3, \dots$; then $\sin(\theta_p(t_n) - \theta(t_n)) = 0$ and $\sin(\theta_p(t_{n+1}) - \theta(t_{n+1})) = 0$. Assume that $\theta_p(t_n) = \theta(t_n)$ and $0 < \theta_p(t) - \theta(t) < \pi$ on the interval (t_n, t_{n+1}) . Then (9.3) guarantees that $\dot{\theta} > 0$ on (t_n, t_{n+1}) and thus θ is increasing on (t_n, t_{n+1}) . Further, we have that either $\theta_p(t_{n+1}) = \theta(t_{n+1})$ or $\theta_p(t_{n+1}) = \theta(t_{n+1}) + \pi$. But since θ is increasing on the interval (t_n, t_{n+1}) , the inequality (9.4) implies that

$$0 \leq \theta_p(t_{n+1}) - \theta(t_{n+1}) = \theta_p(t_{n+1}) - \theta_p(t_n) + \theta(t_n) - \theta(t_{n+1}) < \theta_p(t_{n+1}) - \theta_p(t_n) \leq \frac{\pi}{2},$$

and therefore $\theta_p(t_{n+1}) = \theta(t_{n+1})$. The same conclusion holds if we assume that $-\pi < \theta_p(t) - \theta(t) < 0$ on the interval (t_n, t_{n+1}) , except that in this case θ is monotone decreasing on (t_n, t_{n+1}) . It follows by an induction argument that θ and θ_p have the same values at t_n and θ is monotone on (t_n, t_{n+1}) for every $n = 1, 2, 3, \dots$. The fact that $\lim_{t \rightarrow t_0^-} \theta = \theta_p(t_0)$ is then a simple consequence of continuity of θ_p on $[t_0 - \Delta t, t_0]$. This, in particular, implies (9.1).

Appendix 2. Here we derive (5.7), that is, given

$$(9.5) \quad \Phi(\alpha, x) = \prod_{j=1}^{\lfloor x \rfloor - 1} \left| 1 - \frac{\alpha}{1 - \frac{j}{x}} \right| = \prod_{j=1}^{\lfloor x \rfloor - 1} \left| \frac{x - j - \alpha x}{x - j} \right|,$$

where x is large and $0 < \alpha < 1$, we show that

$$(9.6) \quad \Phi(\alpha, x) = \frac{1}{\pi} \frac{\Gamma(1 + x - \lfloor x \rfloor) \Gamma((1 - \alpha)x) \Gamma(\lfloor x \rfloor - x + \alpha x)}{\Gamma(x)} \sin \pi((1 - \alpha)x - \lfloor (1 - \alpha)x \rfloor).$$

First, given $\lambda > 0$, the relationship

$$(9.7) \quad \frac{\Gamma(m + \lambda + 1)}{\Gamma(l + \lambda)} = \prod_{i=l}^m (i + \lambda)$$

holds for any $l, m \in \mathbb{N}$. By changing the index, $j \rightarrow \lfloor x \rfloor - j$, we have that

$$(9.8) \quad \begin{aligned} \Phi(\alpha, x) &= \prod_{j=1}^{\lfloor x \rfloor - 1} \left| \frac{x - \lfloor x \rfloor - \alpha x + j}{x - \lfloor x \rfloor + j} \right| \\ &= \frac{\prod_{j=1}^{\lfloor x \rfloor - \lfloor (1 - \alpha)x \rfloor - 1} (x - \lfloor x \rfloor - \alpha x + j) \prod_{j=\lfloor x \rfloor - \lfloor (1 - \alpha)x \rfloor}^{\lfloor x \rfloor - 1} (x - \lfloor x \rfloor - \alpha x + j)}{\prod_{j=1}^{\lfloor x \rfloor - 1} (x - \lfloor x \rfloor + j)} \\ &= \frac{\Phi_1(\alpha, x) \Phi_2(\alpha, x)}{\Phi_3(\alpha, x)}. \end{aligned}$$

We consider Φ_1 , Φ_2 , and Φ_3 separately. Using (9.7), we immediately obtain

$$\Phi_3(\alpha, x) = \prod_{j=1}^{\lfloor x \rfloor - 1} (x - \lfloor x \rfloor + j) = \frac{\Gamma(x)}{\Gamma(1 + x - \lfloor x \rfloor)}.$$

By changing the index $j \rightarrow j - [x] + [(1 - \alpha)x]$, we find

$$\Phi_1(\alpha, x) = \prod_{j=0}^{\lfloor(1-\alpha)x\rfloor-1} ((1 - \alpha)x - \lfloor(1 - \alpha)x\rfloor + j) = \frac{\Gamma((1 - \alpha)x)}{\Gamma((1 - \alpha)x - \lfloor(1 - \alpha)x\rfloor)},$$

and by changing the index $j \rightarrow [x] - \lfloor(1 - \alpha)x\rfloor - 1 - j$, we determine that

$$\Phi_2(\alpha, x) = \prod_{j=0}^{\lfloor x\rfloor - \lfloor(1-\alpha)x\rfloor - 1} (\lceil(1 - \alpha)x\rceil - (1 - \alpha)x + j) = \frac{\Gamma(\lfloor x\rfloor - x + \alpha x)}{\Gamma(\lceil(1 - \alpha)x\rceil - (1 - \alpha)x)}.$$

It follows by the properties of the Γ -function that

$$\begin{aligned} \Phi_1(\alpha, x)\Phi_2(\alpha, x) &= \frac{\Gamma((1 - \alpha)x)\Gamma(\lfloor x\rfloor - x + \alpha x)}{\Gamma((1 - \alpha)x - \lfloor(1 - \alpha)x\rfloor)\Gamma(\lceil(1 - \alpha)x\rceil - (1 - \alpha)x)} \\ &= \frac{\Gamma((1 - \alpha)x)\Gamma(\lfloor x\rfloor - x + \alpha x)}{\Gamma((1 - \alpha)x - \lfloor(1 - \alpha)x\rfloor)\Gamma(1 - ((1 - \alpha)x - \lfloor(1 - \alpha)x\rfloor))} \\ &= \frac{1}{\pi}\Gamma((1 - \alpha)x)\Gamma(\lfloor x\rfloor - x + \alpha x)\sin \pi((1 - \alpha)x - \lfloor(1 - \alpha)x\rfloor). \end{aligned}$$

Dividing this expression by Φ_3 , we recover (9.5).

REFERENCES

- [1] W. SMITH AND J. HASHEMI, *Foundations of Materials Science and Engineering*, McGraw-Hill, New York, 2003.
- [2] K. BARMAK, E. EGGELING, R. SHARP, S. ROBERTS, T. SHYU, T. SUN, B. YAO, S. TA'ASAN, D. KINDERLEHRER, A. D. ROLLETT, AND K. COFFEY, *Grain growth and the puzzle of its stagnation in thin films: A detailed comparison of experiments and simulations*, Materials Science Forum, 715-716 (2012), pp. 473-479.
- [3] R. L. FULLMAN, *Metal Interfaces*, American Society for Metals, Metals Park, OH, 1952.
- [4] H. J. FROST, C. V. THOMPSON, C. L. HOWE, AND J. H. WHANG, *A two-dimensional computer simulation of capillarity-driven grain-growth preliminary results*, Scripta Metall., 22 (1988), pp. 65-70.
- [5] T. NAGAI, S. OHTA, K. KAWASAKI, AND T. OKUZONO, *Computer simulation of cellular pattern growth in two and three dimensions*, Phase Transitions, 28 (1990), pp. 177-211.
- [6] R. HENSELER, B. NIETHAMMER, AND F. OTTO, *A reduced model for simulating grain growth*, in Free Boundary Problems, Internat. Ser. Numeri. Math. 147, Springer, New York, 2003, pp. 177-187.
- [7] D. KINDERLEHRER, I. LIVSHITS, G. ROHRER, S. TA'ASAN, AND P. YU, *Mesoscale simulation of the evolution of the grain boundary character distribution*, Materials Science Forum, 467-470 (2004), pp. 1063-1068.
- [8] D. WEYGAND, Y. BRECHET, AND J. LEPINOUX, *A vertex dynamics simulation of grain growth in two dimensions*, Phil. Mag. B, 78 (1998), pp. 329-352.
- [9] D. WEYGAND, Y. BRECHET, AND J. LEPINOUX, *Mechanisms and kinetics of recrystallisation: A two dimensional vertex dynamics simulation*, Interface Sci., 9 (2001), pp. 311-317.
- [10] L. BARRALES-MORA, *2D vertex modeling for the simulation of grain growth and related phenomena*, in Multiscale Modeling of Moving Interfaces in Materials, Math. Comput. Simul. 80, Elsevier, Amsterdam, 2010, pp. 1411-1427.
- [11] M. SYHA AND D. WEYGAND, *A generalized vertex dynamics model for grain growth in three dimensions*, Model. Simul. Mater. Sci. Eng., 18 (2010), 015010.
- [12] D. WEAIRE AND J. P. KERMODE, *Computer simulation of a two-dimensional soap froth I. Method and motivation*, Phil. Mag. B, 48 (1983), pp. 245-259.
- [13] D. WEAIRE AND J. P. KERMODE, *Computer simulation of a two-dimensional soap froth II. Analysis of results*, Phil. Mag. B, 50 (1984), pp. 379-395.
- [14] J.-M. ZHANG, K.-W. XU, AND V. JI, *Experiment and simulation of grain growth in a bidimensional polycrystalline film*, Appl. Surface Sci., 218 (2003), pp. 268-275.

- [15] C. SINCLAIR, D. WEYGAND, J. LEPINOUX, AND Y. BRECHET, *Simulating the topology of recrystallisation in stabilized ferritic stainless steels*, Materials Science Forum, 467–470 (2004), pp. 671–676.
- [16] D. WEYGAND, J. LEPINOUX, AND Y. BRECHET, *On the nucleation of abnormal grain growth: A 2d vertex simulation*, Materials Science Forum, 467–470 (2004), pp. 1123–1128.
- [17] W. W. MULLINS, *The statistical self-similarity hypothesis in grain growth and particle coarsening*, J. Appl. Phys., 59 (1986), pp. 1341–1349.
- [18] A. HARUN, E. A. HOLM, M. P. CLODE, AND M. A. MIODOWNIK, *On computer simulation methods to model Zener pinning*, Acta Materialia, 54 (2006), pp. 3261–3273.
- [19] D. MOLODOV, *Microstructural Design of Advanced Engineering Materials*, Wiley, New York, 2013.
- [20] K. BARMAK, M. EMELIANENKO, D. GOLOVATY, D. KINDERLEHRER, AND S. TA'ASAN, *A new perspective on texture evolution*, Intl. J. Numer. Anal. Model., 5 (2008), pp. 93–108.
- [21] K. BARMAK, M. EMELIANENKO, D. GOLOVATY, D. KINDERLEHRER, AND S. TA'ASAN, *Towards a statistical theory of texture evolution in polycrystals*, SIAM J. Sci. Comput., 30 (2008), pp. 3150–3169.
- [22] K. BARMAK, E. EGGELING, M. EMELIANENKO, Y. EPSHTEYN, D. KINDERLEHRER, R. SHARP, AND S. TA'ASAN, *Critical events, entropy, and the grain boundary character distribution*, Phys. Rev. B, 83 (2011), 134117.
- [23] W. MULLINS, *Solid Surface Morphologies Governed by Capillarity*, in Metal Surfaces: Structure, Energetics and Kinetics, ASM, Cleveland, OH, 1963, pp. 17–66.
- [24] W. MULLINS, *2-Dimensional motion of idealized grain growth*, J. Appl. Phys., 27 (1956), pp. 900–904.
- [25] C. HERRING, *The use of classical macroscopic concepts in surface energy problems*, in Structure and Properties of Solid Surfaces, University of Chicago Press, 1953, pp. 5–81.
- [26] D. KINDERLEHRER AND C. LIU, *Evolution of grain boundaries*, Math. Models Methods Appl. Sci., 11 (2001), pp. 713–729.
- [27] S. ESEDOGLU AND F. OTTO, *Threshold dynamics for networks with arbitrary surface tensions*, Comm. Pure Appl. Math., 68 (2015), pp. 808–864.
- [28] C. V. THOMPSON, *Grain growth and evolution of other cellular structures: III. Grain growth in two dimensions*, in Solid State Physics: Advances in Research and Applications, Academic Press, New York, 2001, pp. 272–278.